<span id="page-1-0"></span>further enhance transition strength. From these considerations, we identified  $CuTaS<sub>3</sub>$  as a potentially strong absorber and set out to assess its optical and electronic properties computationally and then to confirm them experimentally.

CuTaS<sub>3</sub> also brings advantages of low toxicity,<sup>19</sup> material abundance, $20$  and low cost. $21$  Prior reports have focused only on its preparation and crystal structure.<sup>1,2</sup>

Electronic Structure Calculations were performed via density functional theory (DFT) and the plane-wave<br>projector-augmented wave (PAW) method,<sup>22,23</sup> as implemented in the VASP code.<sup>24</sup> The atomic positions were relaxed by the Heyd− Scuseria−Ernzerh hybrid functional (HSE0[6\)](#page-4-0)<sup>25</sup> until all atomic forces were less [tha](#page-4-0)n 0.02 eV Å<sup>−</sup><sup>1</sup> , with latti[ce](#page-4-0) parameters fixed to experimental values. The band structure [an](#page-4-0)d optical absorption coefficients were calculated with the many-body GW approximation<sup>26</sup> perturbatively on the top of wave functions and energy eigenvalues

Consequently, atomic charges are formally  $Cu^{1+}$ ,  $Ta^{5+}$ , and  $S^{2-}$ , consistent with a Cu  $^{10}$  valence band and a Ta  $^{0}$  conduction band. In contrast, Sunshine and Ibers<sup>2</sup> proposed that the valence states were likely to be  $Cu^{2+}Ta^{4+}S_3$ , similar to  $Cu<sub>0.8</sub>Ta<sub>2</sub>S<sub>6</sub>$ . They accounted for the [l](#page-3-0)ow conductivity by assuming the unpaired electrons of Cu and Ta are sequestered in a stro[ng](#page-4-0) Cu−Ta bond. Recent X-ray photoelectron spectroscopy  $(XPS)^1$  and magnetic susceptibility<sup>31</sup> results, however, are consistent with our DFT calculations and assignment of forma[l](#page-3-0) oxidation states:  $Cu^{1+}Ta^{5+}S_3$ .

 $\lambda$  a. We evaluated optical properties by analysis of ground single crystals and thin films. Figure 4 shows a diffuse reflection spectrum of the ground crystals, revealing a band gap near 1 eV.

<span id="page-2-0"></span>R−Γ−Y represents dispersion along [010], i.e., the direction of the one-dimensional Cu- and Ta-centered chains where orbital overlap is significant.

Figure 3 shows total and projected densities of states. A mixture of Cu  $3$  and S  $3$  orbitals characterizes the valence

> XRD analysis confirmed deposition of a crystalline  $CuTaS<sub>3</sub>$ film (Figure S1). The band gap and absorption coefficients were assessed from UV−vis−NIR transmittance (T) and reflect[ance \(R\) s](http://pubs.acs.org/doi/suppl/10.1021/acs.chemmater.6b04730/suppl_file/cm6b04730_si_001.pdf)pectra. Absorption coefficients were calculated with eq  $1,^{32}$

band, while Ta 5 dominates the conduction band. Hence, the electronic structure fulfills the design principle for states from two different elements, each prevailing near VBM and CBM. The spikes in the densities of states reflect the flat-band portions of the band diagram and one-dimensional character of the crystal structure. Based on the total number of electrons available, the valence band is filled, the conduction band empty.

<span id="page-3-0"></span>researchers did not explicitly describe their contacts.<sup>2</sup> We did not measure the resistivity,  $\alpha$ <sub>ac</sub>, in the  $\alpha$ <sub>c</sub> plane. The resistivity of pellets,  $_{\text{p}}$  = 0.4 M $\Omega$  cm, prepared from ground crystals, however, is higher than  $\,$  ,  $\,$   $_{\rm p}$  represents and average of  $\,$   $_{\rm b}$  and  $_{\rm ac}$   $_{\rm ac}$  is anticipated to be higher than  $_{\rm b}$  based on the anisotropic nature of the crystal structure (Figure 1) and the band dispersion (Figure 3). Significant dispersion along [010] contributes to low in the single crystal[, while in](#page-1-0) pressed pellets, contributi[ons from](#page-2-0) the flat, off-axis dispersion in the  $\mu$ plane raises .

The negative Seebeck coefficient,  $-1650$  V K<sup>-1</sup>, of the pellets indicates electrons are majority carriers. This . -type behavior likely arises from sulfur vacancies or Ta-on-Cu antisites  $(Ta_{Cu})$  defects, which place excess electrons in the Ta cry4-672m7.1ri3(s1Tf21.63390T0891650)Tjp1Tf.45920TD3